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| (21) International Application Number: PCT/US (22) International Filing Date: 18 March 1999 ((30) Priority Data: 60/078,470 18 March 1998 (18.03.98) (71) Applicant (for all designated States except US): PROPERTIES INC. [US/US]; Suite 410, 7366 N coln Avenue, Lincolnwood, IL 60646 (US). (72) Inventors; and (15) Inventors/Applicants (for US only): BESSETTE, M. [US/US]; 416 Chaffeeville Road, Storrs, CT 062 SETHUMADHAVAN, Murali [IN/US]; 12 Redc. Framingham, MA 01701 (US). (74) Agent: REIMER, Leah, M.; Cantor Colburn LLP, 88 Road, Windsor, CT 06095 (US). | 99/0594 18.03.9 WORL orth Li ichael, 1 | BY, CA, CH, CN, CU, CZ, DE, DK, EE, ES, FI, GB, GE, GH, GM, HR, HU, ID, IL, IN, IS, JP, KE, KG, KP, KE, KZ, LC, LK, LR, LS, LT, LU, LV, MD, MG, MK, MN, MW, MX, NO, NZ, PL, PT, RO, RU, SD, SE, SG, SI, SK, SL, TJ, TM, TR, TT, UA, UG, US, UZ, VN, YU, ZW, ARIPO patent (GH, GM, KE, LS, MW, SD, SL, SZ, UC, ZW), Eurasian patent (AM, AZ, BY, KG, KZ, MD, RU, T, TM), European patent (AT, BE, CH, CY, DE, DK, ES, F, FR, GB, GR, IE, IT, LU, MC, NL, PT, SE), OAPI pater (BF, BJ, CF, CG, CI, CM, GA, GN, GW, ML, MR, NE, SN, TD, TG). Published With international search report. Before the expiration of the time limit for amending the claims and to be republished in the event of the receipt of amendments. |
| (57) Abstract A low density, microcellular thermoplastic elastomer | ic foam pressure | ER FOAMS AND METHOD OF MANUFACTURE THEREOF is presented, wherein the foam has closed cells. The foam is manufacture drop rate across the die is in the range from about 0.1 to about 4.8 GF from about 320–360 °C. |

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MICROCELLULAR THERMOPLASTIC ELASTOMER FOAMS AND METHOD OF MANUFACTURE THEREOF

Background of the Invention

1. Field of the Invention

This invention relates generally to the field of thermoplastic elastomers. More particularly, this invention relates to microcellular thermoplastic elastomers foamed using supercritical fluids.

2. Description of the Related Art

Thermoplastic elastomers have a wide range of uses, particularly as foams.

Foamed thermoplastic elastomers have found utility as shock absorbing components for equipment casings, automobiles and footwear, as well as in medical applications.

- Despite their popularity, presently available thermoplastic elastomeric foams suffer from various drawbacks and limitations. For example, they are manufactured primarily as open-celled foams, which limits their usefulness in sealing applications. Commonly-used manufacturing processes, requiring chemical or mechanical foaming, are expensive, and yield foams with inherent density limitations. Foams are also not available in the form of thin sheets, i.e., sheets of less than about 50, and preferably less
- available in the form of thin sheets, i.e., sheets of less than about 50, and preferably less than about 15 mils.

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Accordingly, there remains a need in the art for foamed thermoplastic elastomers in the form of thin sheets, that have closed cells and lower densities of less than about 20 lbs/ft³ (pcf), and that are manufactured by economical processes.

Summary of the Invention

The above-described and other problems and deficiencies of the prior art are overcome or alleviated by the present foamed thermoplastic elastomers and method for manufacturing the same, wherein a thermoplastic elastomer resin is foamed using a supercritical fluid under conditions of controlled temperature and pressure drop rate, thereby producing closed-cell, microcellular foams. Such foams may be produced in the form of thin sheets.

The thermoplastic elastomer in accordance with the present invention is aptly suited for uses requiring low density (less than about 20 pcf) and closed cells. These advantageous properties make it particularly suitable for use as industrial gaskets, seals, cushion insoles for shoes, shock pads, elastomeric springs, and appliance foot pads, providing a commercially attractive additions to present materials. These and other features of the invention will become better understood with reference to the following detailed description, drawings, and appended claims.

Detailed Description of the Invention

In accordance with the present invention, a thermoplastic elastomer resin is foamed using a supercritical fluid under conditions of controlled temperature and pressure drop rate, thereby producing closed-cell, microcellular foams. Such foams may be produced in the form of thin sheets.

Thermoplastic elastomeric resins suitable for use with the present invention include, but are not limited to those known in the art such as homopolymers, copolymers, and blends of polyolefins, polyurethanes, polyamides, polyesters, metallocene plastomers, and the like. Preferred elastomers include styrenic block copolymers and thermoplastic vulcanizates (i.e., blends of polyolefin and vulcanized rubber). One particularly preferred elastomer is styrene-ethylene butylene-styrene

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(SEBS), which is available under the trade name KRATON from Shell Chemical Corp. Another particularly preferred elastomer is a thermoplastic vulcanizate having a Shore A hardness of 73 or greater, which is commercially available from Advanced Elastomer Systems, Akron, OH under the trademark SANTOPRENE®. SANTOPRENE® is a blend of polypropylene and ethylene propylene diene (EPDM) copolymer.

Additional components of the foams include, but are not limited to those known in the art, for example nucleating agents, pigments such as carbon black, fire retardants, and/or diffusivity modifiers such as glyceryl monostearate.

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The resin or resin mixtures with the above additives are foamed with a tandem extruder using a supercritical fluid as a blowing agent in accordance with the methods described in U.S. Patent No. 5,158,986 to Cha et al., U.S. Patent Application Serial No. 08/692,060, filed August 2, 1996 as a re-issue of U.S. Patent No. 5,334,06 to Balswin, et al., and U.S. Patent Application Serial No.08/782,955, filed January 13, 1997 to Park et al., all of which are incorporated herein by reference in their entirety. Thus, the polymer and the blowing agent are mixed in the melt stage in the first extruder, and then transferred to the second extruder the temperature and pressure are reduced in order to initiate foaming. The supercritical fluid is preferably carbon dioxide. Alternatively, the polymers and any optional fillers are first melt compounded in highshear equipment such as a BRABENDERTM or twin-screw extruder or other suitable equipment at appropriate temperatures (melt temperature being the minimum and the decomposition temperature of the elastomer being the maximum usable limits). The compounding is preferably performed long enough to obtain a homogeneous mixture on a microscopic level, the homogeneity being measured by the thermogravimetric analysis (for example 15 minutes in a BRABENDER™ and 3 minutes in a twin-screw extruder). This mixture is then mixed with blowing agent as described above.

A number of parameters are important in obtaining the microcellular, closed-cell, low density foams of the present invention, particularly in the form of thin sheets. It has surprisingly been found that obtaining thin sheets of closed cell, microcellular thermoplastic foams depends critically on the melt temperature of the resin and the pressure drop rate across the die. In particular, it has surprisingly been found that an

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increase in the melt temperature of will result in a decrease in the density of the resultant foam. However, the inventors hereof have found that there is a maximum to the temperature increase, such that low density foams are obtained at intermediate temperatures. The preferred melt temperature is in the range of from about 320-360 °F. For thermoplastic vulcanizates such as SANTOPRENE®, the optimal melt temperature is in the range from about 330-360 °F, while for an SEBS such as Kraton, the optimal melt temperature is in the range from about 320-350 °F.

Importantly, the pressure drop rate across the die must be maintained at an intermediate level. Ordinarily, increasing the pressure drop rate across the die (dp/dt) results in a decrease in the density of the obtained foams, as well as very fine cell structure. However, it has been found that the optimal dp/dt has a maximum value that is preferably not exceeded. Thus, higher pressure drop rates in the range from 11-15 GPA (die of 0.042 inches tapered) gives rise to low density foams of about 8 pcf with uniform cell sizes of about 150 microns for the thermoplastic vulcanizate SANTOPRENE® 201-64, and low density foams of about 6 pcf and non-uniform cell size of about 250 microns for Kraton GLS 7715-9. Both foams had very poor mechanical strength.

Lower pressure drop rates in the range from about 0.1 to about 0.8 GPA (die of 0.08 x 1.2 inches) gave rise to higher density foams of about of about 14 pcf with uniform cell sizes of about 150 microns for the thermoplastic vulcanizate SANTOPRENE® 201-64, and low density foams of about 8 pcf and non-uniform cell size of about 250 microns for Kraton GLS 7715-9. Both foams had fairly good mechanical properties.

However, intermediate pressure drop rates in the range from about 0.8 to about 2.6 GPa (die of 0.06 x 1.2 inches) gave rise to foams with a density of about of about 12 pcf with uniform cell sizes of about 100 microns for the thermoplastic vulcanizate SANTOPRENE® 201-64 low density foams of about 9 pcf with uniform cell size for SANTOPRENE® 201-73. Intermediate pressure drop rates in the range from about 1.8 to about 4.8 GPa (die of 0.05 x 0.75 inches) gave rise to higher density foams of about 12 pcf (un-optimized) and uniform cell size of about 150 microns for Kraton GLS

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7715-9. All of these foams had very good mechanical properties. Accordingly, pressure drop rates in the range form about 0.1 to about 4.8 GPa are preferred in the practice of the present invention, while pressure drop rates in the range from about 0.8 to about 4.8 GPa are most preferred.

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Other parameters which affect the production and properties of the foams of the present invention include the concentration of the blowing agent (supercritical fluid), which affects the density of the foam. Optimal blowing agent concentration appears to be about 3% by weight of the polymer. Beyond 6%, gas comes out of solution and the polymer cannot support itself. Cooling the foam as it came out of the die resulted in a density reduction of at least about 15%.

Melt strength and viscosity of the foaming composition are other important parameters in successfully producing foam sheets. Melt strength in particular is improved by the addition of one or both of a high molecular weight ethylene propylene diene terpolymer (EPDM) and a low density polyethylene (LDPE). Suitable high molecular weight EPDM terpolymers include but are not limited to those having diene monomers such as dicyclopentadiene, 1,4-hexadiene or ethylidene norbornene. A higher molecular weight elastomer comprising ethylidene norbornene is commercially available under the trade name Royalene 505 from the Uniroyal Chemical Company. Another suitable high molecular weight EPDM is available from Uniroyal under the trade name IM-7100. Suitable LDPE's include those available from under the trade names D4032F from Eastman Chemical Co., DFOB6005 from Union Carbide Corp., and 303AA and 601AA from Westlake Polymers.

The above-described parameters such as melt temperature, melt strength, viscosity, pressure drop rate, and blowing agent concentration may be adjusted as required in order to optimize the density, cell structure and size, compression set, tensile strength and elongation, thickness of a sheet, and skin of the foam. The optimal density is less than about 20 pcf, preferably less than about 15 pcf, more preferably less than about 10 pcf, and most preferably less than about 3 pcf. The optimal sheet thickness in the range of about 0.01 to about 0.08 inches, preferably about 0.015 about 0.05 inches. The optimal compression set is less than about 30%, and preferably less

than about 20%. Sheets will be at least about one inch wide, preferably at least about 8 inches wide, more preferably at least about 18 inches wide, even more preferably at least about 24 inches wide, and most preferably at least about 36 inches wide.

The following non-limiting examples further describe the document feed component of the present invention.

Examples

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Procedures

Materials were obtained from the sources listed in Table 1 below. Table 1.

| 10 | Trade Name | Source | Description |
|------------|------------------------|-------------------------|--------------------------------------------------|
| • | SANTOPRENE® 201- 64 | American Elastomer | Thermoplastic vulcanizate resin, Shore A = 64 |
| | SANTOPRENE® 201- 73 | American Elastomer | Thermoplastic vulcanizate resin, Shore A = 73 |
| 15 | KRATON GLS 7715-9 | Shell Chemical Corp. | SEBS resin |
| | PRO-FAX 6823. | Montell N. Am. | High molecular weight polypropylene |
| | PRO-FAX 814 | Montell N. Am. | High molecular weight polypropylene |
| | - | Bakaert Corp | Stainless steel filler |
| 20 | D4032F | Eastman Chemical | low density polyethylene |
| | DFOB60005 | Union Carbide | low density polyethylene |
| | 303AA | Westlake Polymers | low density polyethylene |
| 2 5 | 601AA | Westlake Polymers | low density polyethylene |

The following samples (Table 2) were formulated using a tandem extruder having an output rate of 100 lbs/hour, which may be varied by adjustment the revolutions per minute of the screw.

| Table 2 | | | | | | | |
|--------------------|-----------|--------------|------------|--------|----------|--------------|------------|
| | Head | Melt | Blowing | | | Average cell | |
| Polymer . | Pressure, | Temperature, | Agent, % | dp/dt, | Density, | diameter, | Mechanical |
| | psi | E | by weight | GPA/s | pcf | microns | properties |
| | | | of polymer | | | | |
| SANTOPRENE® 201-64 | 1420 | 370 | 1 | 0.5 | 39 | 80 | Cood* |
| SANTOPRENE 201-64 | 1860 | 340 | 1.9 | 0.7 | 18 | 120 | Good* |
| SANTOPRENE 201-64 | 1340 | 370 | 1.9 | 0.5 | 22 | 70 | Bad skin |
| SANTOPRENE 201-64 | 1040 | 342 | 2.9 | 0.1 | 16 | 120 | Good* |
| SANTOPRENE 201-64 | 1670 | 331 | 4.1 | 0.7 | 13 | 120 | Good* |
| SANTOPRENE 201-64 | 2190 | 338 | 3.7 | 1.1 | 13 | 80 | Poor**** |
| SANTOPRENE 201-64 | 2110 | 352 | 4.7 | 12.4 | 12 | . 20 | Good* |
| SANTOPRENE 201-73 | 2990 | 332 | 2.7 | 1.4 | 6 | 80 | Good* |
| Kraton GLS 7715-9 | 1560 | 330 | 2 | 0.6 | 11 | 250 | Good** |
| Kraton GLS 7715-9 | 1060 | 351 | 2 | 0.5 | 17 | 250 | Good** |
| Kraton GLS 7715-9 | 1200 | 322 | 4 | 0.5 | 9 | 250 | Good** |
| Kraton GLS 7715-9 | 1550 | 340 | ۳. | 7.2 | 9. | 250 | Poor*** |
| Kraton GLS 7715-9 | 2140 | 342 | 2.6 | 1.9 | 10 | 150 | 370 |

*Compression set resistance about 15%, good skin, tensile strength at least about 120 psi, elongation break at least about 60%.

**Compression set resistance about 25-35%, good skin, tensile strength at least about 60 psi, elongation break at least about

***Compression set resistance about 25-35%, good skin, tensile strength at least about 100 psi, elongation break at least about

****Elongation break less than about 10%.

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In order to optimize viscosity and melt strength, different elastomer blends were prepared in accordance with Table 3 below (parts by weight of the total composition). Viscosity was calculated based on the torque values observed while blending in a Brabender mixer. Overall, blends of SANTOPRENE® 201-73 with high molecular weight polypropylene and EPDM had higher viscosities than SANTOPRENE® 201-73 alone. Blends of SANTOPRENE® 201-73 with high molecular weight polypropylene and EPDM (6:1:3) had higher viscosities than the same components in ratios of 6:2:2. Blends comprising high molecular weight polypropylene (PP 6823) were more viscous than blends comprising high melt strength polypropylene (PP 814).

Based on the viscosity (and other) results, the compositions set forth in Table 2 were blended and tested for melt strength. In Table 2, S 73 is SANTOPRENE® 201-73, S 64 is SANTOPRENE® 201-73, PP 6823 is PRO-FAX 6823, PP 814 is PRO-FAX 814, SS is stainless steel filler available from Bakaert Corp., and LDPE is the low density polyethylene having the trade designation D4032F unless otherwise noted. Three different tests were used in order to provide and indication of melt strength.

DMTA (dynamic mechanical thermal analysis) yields the response of a material to a periodic stress at various temperatures (storage modulus), which is directly proportional to elastic modulus. Thus, samples were blended in a Brabender, hot pressed at 380 °F, and allowed to rest for at least 24 hours before testing. Average dimensions of each sample were approximately 20 x 4 x 1 mm. Storage modulus of each sample was measured over the temperature range of 50 to 160 °C using a sweep rate of 5 °C per minute. The advantages of this test are that is very sensitive to molecular motion and it is conducted across varying temperatures. Generally, as shown in Table 3 below, LDPE blends show low modulus at high temperature, and PP 6823 blends show higher moduli than PP 814 blends. Samples comprising 20% SS fibers show result comparable to those without (data not shown).

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Tensile tests measure buildup of force (or stress) as the specimen is stretched at a constant rate. It yields force (and thus strength) values, and the results are close to those of the melt test at the melt temperature of the material. Samples having an average size of 0.6 x 0.5 x 0.1 mm were prepared as in the DMTA test, wherein the first value (0.6) is the gauge length. Tensile strength is measured with a Model 1101 Instron using a 100-pound load cell and a cross-head speed of 20 inches per minute at 165 °C. As measured by this test, compositions comprising SANTOPRENE, EPDM and PP 6823 showed higher elongation and toughness. Samples comprising 20% SS fibers show result comparable to those without (data not shown).

A near melt extensiometer was also used to provide an indication of melt strength (NME). Freshly extruded polymer in the shape of a strand or rod is extended between tow pairs of moving rollers to measure force and elongation at break. Again, compositions comprising SANTOPRENE, EPDM and PP 6823 showed higher elongation and toughness.

Results from each of these three tests is presented in Table 3 below.

| | | ٠ | |
|---|---|---|--|
| ¢ | * | 3 | |
| | q | د | |
| • | ć | 3 | |
| _ | c | 9 | |
| t | | 7 | |

| | Resi | Resin formulation (ratio) | tion (rat | (oi | | | | DMT/ | \ (Storage | DMTA (Storage Modulus, MPa) | MPa) | | Ter | Tensile | N | NME |
|------|----------|---------------------------------------------|-----------|----------|----------|----|--------|--------|------------|-----------------------------|--------|----------|---------|---------|---------|-----------|
| S 73 | S 64 | EPDM | Вď | PP | LDPE | SS | 120 °C | J. 0£1 | 140 °C | J. 051 | J. 091 | Jes °C | Max. | Мах. | Мах. | Strain at |
| | | | 6823 | 814 | | | | | | | | | Stress, | Strain, | Stress, | Break, |
| | | | | · ··· | | | | | | | | | Jb/in² | % | lb/in² | % |
| 901 | ; | : | ; | : | : | : | 9.47 | 8.10 | 6.49 | 4.34 | 1.83 | 10:1 | 110.4 | 627.8 | 35 | 300 |
| 08 | ; | ; | ' | 20 | : | : | 27.37 | 23.07 | 18.43 | 12.33 | 3.13 | 0.77 | 95.5 | 1045 | 29 | 350 |
| 09 | : | 30 | ; | 2 | ı | ; | 8.72 | 7.51 | 6.15 | 4.28 | 1.99 | 1.09 | ; | | : | 1 |
| 09 | ; | 30 | 0 | 1 | | • | 9.25 | 8.09 | 6.83 | 5.30 | 2.92 | 1.62 | 36 | 1400 | 50 | 360 |
| 09 | ; | 20 | 1 | 20 | | 1. | 16.74 | 14.51 | 12.13 | 8.89 | 2.86 | 1.03 | : | ; | 1 | ; |
| 09 | ; | 20 | 20 | ; | : | : | 20.84 | 17.74 | 14.53 | 10.66 | 5.32 | 2.31 | 65.5 | 1600 | 50 | 350 |
| 08 | 1 | : | 20 | ; | ł | 1 | 29.56 | 24.72 | 19.58 | 13.63 | 5.49 | 2.08 | 136 | 1004 | 48 | 310 |
| 08 | 1 | 20 | : | : | : : | : | 4.38 | 4.08 | 3.57 | 2.90 | 1.76 | 1.09 | 52 | 813 | 57 | 340 |
| ; | ; | ٠ | ; | 90_ | 1 | 1 | 421.20 | 301.80 | 186.00 | 89.60 | 9.10 | : | : | | : | ı |
| : | : | | 100 | 1 | 1 | 1 | 121.80 | 97.80 | 73.70 | 49.20 | 22.50 | 14.10 | 1 | : | ; | ì |
| | 100 | 1 | , | · : | - | - | t | i | ; | 1 | : | ; | ı | 1 | 33 | 310 |
| • | 99 | 20 | 20 | 1 | 1 | - | 18.14 | 15.32 | 12.59 | 9.42 | 4.46 | 1.89 | 78.5 | 1630 | 99 | 345 |
| | 09 | 70 | : | : | 20* | 1 | 1.64 | 1.35 | 1.18 | 0.97 | 69:0 | 0.59 | 13 | 1220 | 51 | 378 |
| • | 09 | 20 | 1 | - | 20* | : | 1.85 | 1.59 | 1.37 | 1.08 | 0.65 | 0.59 | 15.5 | 602.2 | : | i |
| ; | 06. | 20 | : | | 20** | ; | 2.53 | 2.24 | 1.90 | 1.28 | 0.62 | 0.55 | 10.5 | 360 | 1 | 1 |
| | 8 | 70 | | | 20*** | 1 | 1.90 | 1.67 | 1.47 | 1.15 | 99.0 | 0.59 | = | 55 | ŀ | : |
| 48 | : | 91 . | 16 | : | 1 | 20 | 20.29 | 16.46 | 12.47 | 7.97 | 3.51 | 2.02 | 58 | 1600 | . 65 | 322 |
| | DPE have | *LDPE having the trade designation DFOB6005 | de design | nation L |)FOB6005 | | | | | | | | | | | |

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*LDPE having the trade designation DFOB6005
**LDPE having the trade designation 303AA

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***LDPE having the trade designation 601AA

The foamed thermoplastic elastomer in accordance with the present invention is suited for uses requiring low density (less than about 20 pcf) and closed cells. These advantageous properties make it particularly suitable for use as industrial gaskets, seals, cushion insoles for shoes, shock pads, elastomeric springs, and appliance foot pads, providing a commercially attractive additions to present materials.

While preferred embodiments have been shown and described, various modifications and substitutions may be made thereto without departing from the spirit and scope of the invention. Accordingly, it is understood that the present invention has been described by way of illustrations and not limitation.

10 What is claimed is:

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CLAIM 1. A method of making a thermoplastic elastomeric foam, comprising mixing a thermoplastic elastomeric composition with a supercritical fluid in a first extruder at a first temperature and pressure; and

extruding the mixture across a die having a pressure and temperature change to a second temperature and pressure, thereby causing foaming of the thermoplastic elastomeric composition to form a closed cell, microcellular foamed thermoplastic elastomer.

CLAIM 2. The method of claim 1, wherein

the thermoplastic elastomeric resin composition is mixed with the supercritical fluid in a first extruder and then transferred to a second extruder.

CLAIM 3. The method of claim 1, wherein

the thermoplastic elastomeric resin composition is selected from the group consisting of styrenic copolymers, thermoplastic vulcanizates, styrene-ethylenebutylene-styrene, and a thermoplastic vulcanizate having a Shore A hardness of at least 73.

CLAIM 4. The method of claim 3, wherein

the melt temperature of the thermoplastic elastomeric resin composition is in the range from about 320-360 °F.

CLAIM 5. The method of claim 2, wherein

the melt temperature of the styrene-ethylenebutylene-styrene is in the range from about 320-350 °F, and the melt temperature of the thermoplastic vulcanizate is in the range from about 330-360 °F.

CLAIM 6. The method of claim 3, wherein

the thermoplastic resin composition further comprises an ethylene propylene diene, terpolymer, a low density polyethylene, or a combination thereof.

CLAIM 7. The method of claim 1, wherein

the pressure drop rate across the die is in the range from about 0.1 to about 4.8 GPa.

CLAIM 8. The method of claim 1, wherein

the pressure drop rate across the die is in the range from about 0.8 to about 4.8 GPa.

CLAIM 9. The method of claim 1, wherein

the pressure drop rate across the die is in the range from about 1.4 to about 1.4, and the second temperature is about 330 °C.

CLAIM 10. The method of claim 1 wherein the thermoplastic resin composition is melt compounded prior to mixing with the supercritical fluid.

CLAIM 11. A method of making a thermoplastic elastomeric foam, comprising melt compounding a thermoplastic elastomeric resin composition, wherein the melt temperature of the resin composition is in the range from about 320-360 °F;

mixing the compounded resin composition with a supercritical fluid in a first extruder at a first temperature and pressure; and

adjusting the mixture to a second temperature and pressure, thereby causing foaming of the thermoplastic elastomeric foam to form a closed cell, microcellular foamed thermoplastic elastomer, wherein the pressure drop rate across the die is in the range from about 0.1 to about 4.8 GPa.

CLAIM 12. The method of claim 11, wherein

the pressure drop rate across the die is in the range from about 0.8 to about 4.8 GPa.

CLAIM 13. The method of claim 11, wherein

the thermoplastic elastomeric resin composition is mixed with the supercritical fluid in a first extruder and then transferred to a second extruder.

CLAIM 14. The method of claim 11, wherein

the thermoplastic elastomeric resin composition is selected from the group consisting of styrenic copolymers, thermoplastic vulcanizates; styrene-ethylenebutylene-styrene, and a thermoplastic vulcanizate having a Shore A hardness of at least 73.

CLAIM 15. The method of claim 14, wherein

the thermoplastic resin composition further comprises an ethylene propylene diene, terpolymer, a low density polyethylene, or a combination thereof.

- CLAIM 16. The method of claim 11, wherein the thermoplastic resin composition further comprises a filler.
- CLAIM 17. The method of claim 11, wherein the formed foam is low density and has closed cells in the microcellular range.
- CLAIM 18. A thermoplastic, elastomeric foam manufactured by the method of claim 1.
- CLAIM 19. A thermoplastic, elastomeric foam manufactured by the method of claim 11.

INTERNATIONAL SEARCH REPORT

International application No. PCT/US99/05945

| IPC(6) :C US CL :5 | SIFICATION OF SUBJECT MATTER :08F 14/06, 36/04; C08J 9/04 21/79, 142, 146, 148, 150 International Patent Classification (IPC) or to both r | national classification and IPC | |
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| | cumentation searched (classification system followed | by classification symbols) | |
| U.S. : 52 | 21/79, 142, 146, 148, 150 | | |
| Documentation | on searched other than minimum documentation to the | extent that such documents are included in the fields searched | |
| Electronic dat | ta base consulted during the international search (na | me of data base and, where practicable, search terms used) | |
| c. Docu | IMENTS CONSIDERED TO BE RELEVANT | | |
| Category* | Citation of document, with indication, where app | propriate, of the relevant passages Relevant to claim No. | |
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| Furthe | or documents are listed in the continuation of Box C | . See patent family annex. | |
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